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## The Physical and Thermodynamic Functions of Borides

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In the paper the physical properties and thermodynamic functions of borides  $X_2B$  ( $X=W, Mo, Mn, Fe, Co, Ni$  та  $Cr$ ) are studied with accounting for fluctuation processes. We use the microstructure analysis, the X-ray structural and the durometric analyses to determine the physical properties of alloys. In the paper it is determined the phase composition and physical properties of borides. In this paper for the first time it is determined the thermodynamic functions of borides using the Hillert and Staffansson model with accounting for the first degree approximation of high-temperature expansion for the free energy potential of binary alloys. We obtain the temperature dependences for such thermodynamic functions as Gibbs free energy, entropy, enthalpy and heat capacity  $C_p$  along with their values at the formation temperature for  $X_2B$  ( $X=W, Mo, Mn, Fe, Co, Ni$  та  $Cr$ ). The approach under consideration enables to give more thorough from the thermodynamic point of view description of borides formed from the liquid. The outcomes of the thermodynamic function calculation for borides are in good agreement with experimental data and results of other authors.

**Key words:** borides, Gibbs energy, entropy, enthalpy, heat capacity, fluctuation process

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### Introduction

The  $W_2B$ ,  $Mn_2B$ ,  $Fe_2B$ ,  $Co_2B$ ,  $Ni_2B$ ,  $Mo_2B$  and  $Cr_2B$  borides have a body-centered tetragonal cell with 12 atoms per cell and belong to the structural type of  $CuAl_2$   $D_{4h}^{18} - I4/mcm$  [1-5].

The W-B, Mn-B, Fe-B, Co-B, Ni-B, Mo-B and Cr-B systems were studied both experimentally and theoretically [6-12]. In the Refs. [6-12] the authors give results of calculation for Gibbs energy of borides using the models that can be applied only for equilibrium conditions. The objective of this paper is to study physical properties and thermodynamic functions of borides, their temperature dependences with consideration of the first degree approximation of high-temperature expansion for the free energy potential of binary alloys.

### I. Materials and research technique

The investigation was carried out on specimens with boron content of 5 - 9.5 % (wt.), the rest is metal X ( $X=W, Mo, Mn, Fe, Co, Ni$  та  $Cr$ ), for which is used burden consisting of metal with content 99.99 %, amorphous Boron (with Boron content of 97.5,0 % (wt.)). The smelting of specimens was carried out in Taman furnace with graphite hearth, melting of

specimens was carried out in alundum sagger under argon atmosphere. The cooling rate for alloys amounted to 10 K/s. To ascertain alloy chemistry the chemical and spectrographic analysis were used [13]. The microhardness of various phase components was determined by using a PMT-3.

The phase compositions of the alloys were determined by the method of X-ray spectral microanalysis in a JSM-6490 microscope and with the help of a Neofot-21 optical microscope.

Make use of X-radiation diffraction analysis, which was realized in a DRON-3 diffractometer in the monochromatized Fe-K $\alpha$ -radiation.

### II. Results and discussion

The microstructure of the W-B, Mn-B, Fe-B, Co-B, Ni-B, Mo-B and Cr-B as-cast alloys at boron content over 33.3 % (atomic) contains plane-faced borides (Fig. 1, a, b).

The structure of borides consists of alternating planes of iron atoms (forming the identical square grids) and planes of boron atoms [14]. The occurrence of the planes of different packing density has to enable the sharp growth anisotropy of boride. Anisotropy and interatomic forces result in that boride crystals are right regular square prisms in shape and grow in (011)

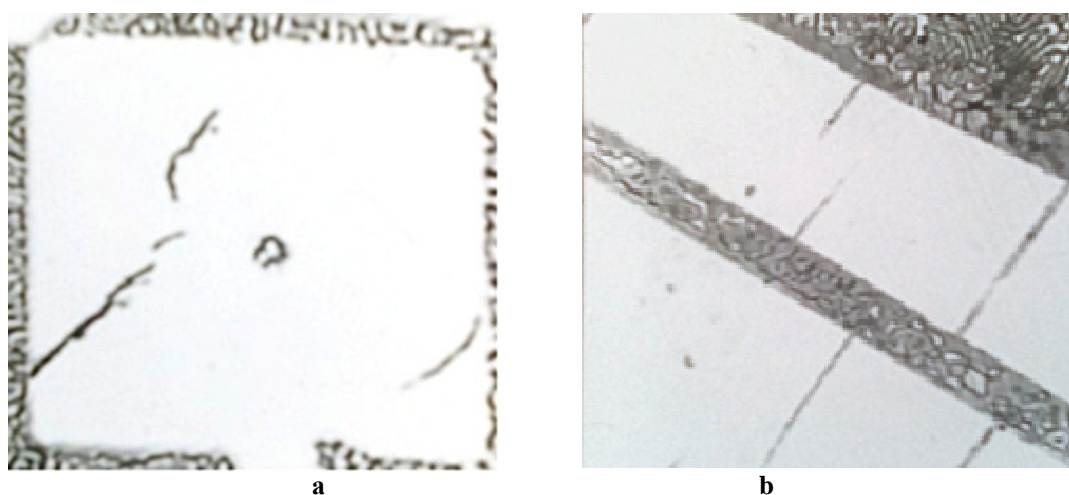


Fig. 1. Microstructure of alloys with boron content of 7 % (w.) Fe-B (a), Co-B (b)

Table 1

Crystal lattice parameters for borides with boron content of 7 % (w.)

| Borides           | Lattice constant, (exp) |        | Lattice constant (tab) |        | Reference |
|-------------------|-------------------------|--------|------------------------|--------|-----------|
|                   | a, Å                    | c, Å   | a, Å                   | c, Å   |           |
| W <sub>2</sub> B  | 5.565                   | 4.741  | 5.564                  | 4.740  | [2]       |
| Mn <sub>2</sub> B | 5.147                   | 4.21   | 5.148                  | 4.208  | [2]       |
| Fe <sub>2</sub> B | 5.1074                  | 4.2464 | 5.109                  | 4.249  | [3]       |
| Co <sub>2</sub> B | 5.016                   | 4.21   | 5.016                  | 4.22   | [3]       |
| Ni <sub>2</sub> B | 5.01                    | 4.246  | 4.993                  | 4.249  | [4]       |
| Mo <sub>2</sub> B | 5.542                   | 4.736  | 5.543                  | 4.735  | [2]       |
| Cr <sub>2</sub> B | 5.1287                  | 4.282  | 5.1283                 | 4.2818 | [5]       |

Table 2

Dependence of size of crystallites L, level of microstresses, density of dislocations  $\rho$  in borides with boron content of 7.0 % (wt.)

| Borides           | Size of Crystallites L, Å | Level of Microstresses M | Density of Dislocations $\rho \times 10^{10}$ , sm <sup>-2</sup> |
|-------------------|---------------------------|--------------------------|--|
| W <sub>2</sub> B  | 856                       | $5.01 \cdot 10^{-3}$     | 8.64   |
| Mn <sub>2</sub> B | 584                       | $2.12 \cdot 10^{-3}$     | 6.79   |
| Fe <sub>2</sub> B | 452                       | $2.13 \cdot 10^{-3}$     | 6.52   |
| Co <sub>2</sub> B | 872                       | $3.85 \cdot 10^{-3}$     | 6.92   |
| Ni <sub>2</sub> B | 692                       | $1.63 \cdot 10^{-3}$     | 7.6  |
| Mo <sub>2</sub> B | 892                       | $3.23 \cdot 10^{-3}$     | 6.87   |
| Cr <sub>2</sub> B | 815                       | $4.04 \cdot 10^{-3}$     | 7.79   |

direction. In this direction the bond is strongest and growth rate of prisms, based on (011) face, leads to their impurity and defect enrichment, which gives rise to inhomogeneities such as pores, cracks and others (Fig. 1, a, b).

Determinations of lattice parameter of the phases by using X-radiation analysis show their correlation to tabular data (Table 1).

For the W<sub>2</sub>B boride we observed a slight increase of microstrain degree and dislocation density compared to other borides (Table 2).

The obtaining of thermodynamic functions values for the W<sub>2</sub>B, Mn<sub>2</sub>B, Fe<sub>2</sub>B, Co<sub>2</sub>B, Ni<sub>2</sub>B, Mo<sub>2</sub>B and Cr<sub>2</sub>B borides is quite difficult. Thereby, accounting for the first degree approximation of high-temperature expansion for the free energy potential in the Hillert and Staffansson model enables to determine the thermodynamic functions of borides and their temperature dependences

theoretically.

### 2.1. Gibbs energy for the W<sub>2</sub>B, Mn<sub>2</sub>B, Fe<sub>2</sub>B, Co<sub>2</sub>B, Ni<sub>2</sub>B, Mo<sub>2</sub>B and Cr<sub>2</sub>B borides.

Gibbs energy of the phase is known to be a function of variables  $G = G(p, T, y)$ , where  $p$  is pressure,  $T$  denotes temperature,  $y$  represents weight content of the elements. For mole fraction of components in compound or alloy the condition  $\sum_{i=1}^2 y_i = 1$  holds.

As it is known, the sublattice model by Hillert and Staffansson [15] enables to calculate Gibbs energies of the phases for equilibrium state. The potentials within the Hillert and Staffansson model don't take an account for the first degree approximation of high-temperature series for binary alloy thermodynamic potential that must be borne in mind when deriving the Gibbs energy for the phases formed from liquid and involving fluctuation

Table 3

Temperature dependence of Gibbs energy and enthalpy value corresponding to the formation of borides

| Borides           | Equation of Gibbs energy                 | $G_m^{MeB}$ ,<br>(Дж/МОЛЬ) | Reference                    | $\Delta H_m^{MeB}$ ,<br>(Дж/МОЛЬ) | Reference                    |
|-------------------|--|----------------------------|------------------------------|-----------------------------------|------------------------------|
| W <sub>2</sub> B  | $-72687 + 5.3T - 2.3 \cdot 10^5 T^{-1}$  | -58872.79                  | First-principles calculation | -83543.1                          | First-principles calculation |
|                   | $-97250 + 1.787T$                        | -413310.24                 | [9]                          | -87000                            | [9]                          |
| Mn <sub>2</sub> B | $-35264 - 0.35T - 8.8 \cdot 10^5 T$      | -36389.38                  | First-principles calculation | -32856.35                         | First-principles calculation |
|                   | $-31254 + 0.293T$                        | -30705.2                   | [27]                         | -38600                            | [34]                         |
|                   | $-316519 - 6.17T$                        | -43208.31                  | [7]                          | -31700                            | [32]                         |
| Fe <sub>2</sub> B | $-29726 + 2.5T - 10^5 T^{-1}$            | -26217.7                   | First-principles calculation | -31630                            | In this paper                |
|                   | $-29365 + 6.54T$                         | -22244.5                   | [28]                         | -24294.83                         | [35]                         |
|                   | $-31000 + 3.37T$                         | -25362                     | [29]                         | -32600.72                         | [36]                         |
| Co <sub>2</sub> B | $-32128 + 2.1T - 2.6 \cdot 10^5 T^{-1}$  | -28989.99                  | First-principles calculation | -25345.82                         | First-principles calculation |
|                   | $-269000 - 1.41T$                        | -29117.9                   | [8]                          | -26400                            | [37]                         |
|                   | $-28564 + 5.077T$                        | -20588.89                  | [30]                         |                                   |                              |
| Ni <sub>2</sub> B | $-35157 + 1.36T - 1.5 \cdot 10^5 T^{-1}$ | -32971.4                   | First-principles calculation | -27363.90                         | First-principles calculation |
|                   | $-22500 + 2.5T$                          | -21500                     | [10]                         | -21129.22                         | [10]                         |
|                   | $-24100,75 + 4,017T$                     | -17380.32                  | [25]                         | -24670                            | [25]                         |
| Mo <sub>2</sub> B | $-50269 + 5.2T + 2.4 \cdot 10^5 T^{-1}$  | -37054.36                  | First-principles calculation | -37895.10                         | First-principles calculation |
|                   | $-42800 + 2.43T$                         | -35775.71                  | [21]                         | -42860                            | [21]                         |
|                   | $-42176 + 2T$                            | -37030                     | [22]                         |                                   |                              |
| Cr <sub>2</sub> B | $-32947 + 0.78T - 1.3 \cdot 10^6 T^{-1}$ | -31514.66                  | First-principles calculation | -42930.90                         | First-principles calculation |
|                   | $-30848 + 1.48T$                         | -28001.96                  | [32]                         | -41417.6                          | [40]                         |
|                   | $-28285 - 0.33T$                         | -28903.09                  | [31]                         |                                   |                              |

processes into consideration. As we know from theory of binary alloys, the partition function of such a system cannot be computed exactly, but according to Kirkwood technique it may be written in the form of infinite series

in powers of 1/T [16-17].

So, we define Gibbs energy for boride with accounting for the first degree approximation as:

$$G_m^{MeB} = y_{Me} {}^0G_{Me} + y_B {}^0G_B + RT(2y_{Me} \ln y_{Me} + y_B \ln y_B) + y_{Me}y_B L_{Me:B} - \frac{L_{Me:B}^2 y_{Me}^2 y_B^2}{2ZRT}, \quad (1)$$

where Z is coordination number, which equals Z=12 for boride [2-3].

Using data for pure components  ${}^0G_{Me}$ ,  ${}^0G_B$  [19-20] and information on interaction energy of components in the phase  $L_{Me:B}$  from the papers [7, 9, 20-26], we obtain temperature dependences of Gibbs energy for borides.

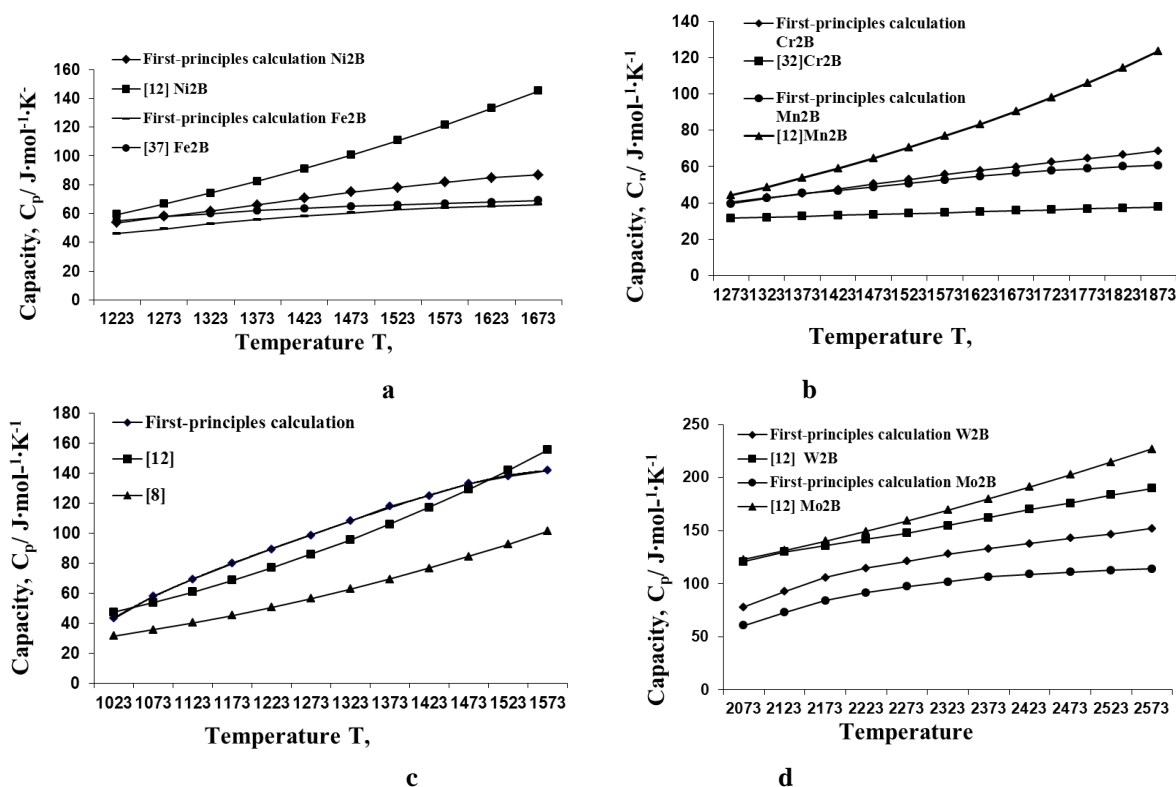
Table 3 contains computational data on Gibbs energy and enthalpy of borides at temperature of their formation.

As we can see from Table 3, obtained values of Gibbs energy are in good agreement with those of other authors.

Thus, obtained temperature dependences of Gibbs energies of borides enable to determine their values in high-temperature region, as well as Gibbs energies of formation of these phases from the liquid.

## 2.2. Entropy, enthalpy and heat capacity $C_p$ for borides.

One of the most important thermodynamic characteristics of the phase is entropy. The entropy of



**Fig. 2.** Temperature dependence of heat capacity  $C_p$  for borides: Fe<sub>2</sub>B and Ni<sub>2</sub>B (a), Cr<sub>2</sub>B and Mn<sub>2</sub>B (b), Co<sub>2</sub>B (c), W<sub>2</sub>B and Mo<sub>2</sub>B (d).

borides one can determine from the formula:

$$S = -\left(\frac{\partial G}{\partial T}\right)_p = -R(y_{Me} \ln y_{Me} + y_B \ln y_B) - \frac{L_{Me:B}}{2ZRT^2} y_{Me}^2 y_B^2$$

The accounting in Gibbs energy for the first degree approximation of high-temperature expansion of thermodynamic potential enables to determine the enthalpy of borides. To calculate the enthalpy of borides we use relation [33]:  $\Delta H = \Delta G + T\Delta S$ .

Temperature dependence of enthalpy of the W<sub>2</sub>B, Mn<sub>2</sub>B, Fe<sub>2</sub>B, Co<sub>2</sub>B, Ni<sub>2</sub>B, Mo<sub>2</sub>B and Cr<sub>2</sub>B phases has the form:

$$H^{W_2B} = -127698 + 21.3T - 3.1 \cdot 10^5 T^{-1},$$

$$H^{Mn_2B} = -35126 + 1.36T - 5.2 \cdot 10^5 T^{-1},$$

$$H^{Fe_2B} = -32527 + 0.5T + 10^5 T^{-1},$$

$$H^{Co_2B} = -26458 + 0.8T - 2.3 \cdot 10^5 T^{-1}$$

$$H^{Ni_2B} = -30208 + 1.7T - 2.45 \cdot 10^5 T^{-1},$$

$$H^{Mo_2B} = -43698 + 2.3T - 3.1 \cdot 10^5 T^{-1}$$

$$H^{Cr_2B} = -48123 + 2.7T - 1.5 \cdot 10^5 T^{-1}$$

Table 3 gives the comparison results of borides' enthalpy values obtained in this paper with those obtained experimentally and by calculation by other authors.

So, the outcomes of this research (Table 3) are in good agreement with those of other authors [10, 21, 25,

32-37].

We calculate heat capacity for the W<sub>2</sub>B, Mn<sub>2</sub>B, Fe<sub>2</sub>B, Co<sub>2</sub>B, Ni<sub>2</sub>B, Mo<sub>2</sub>B and Cr<sub>2</sub>B borides using the formula

$$C_p = T \left( \frac{\partial S}{\partial T} \right)_p = \frac{L_{Fe:B}^2}{RZT^2} \cdot y_{Fe}^2 y_B^2.$$

As result, the obtained temperature dependence of heat capacity  $C_p$  for borides (Fig. 2) matches with data of Refs. [12, 33, 37, 40].

Judge by the results obtained we can conclude that accounting for contribution of the first degree approximation of high-temperature expansion of the free energy potential for binary alloys in the Hillert and Staffansson models enables to calculate such thermodynamic quantities of the W<sub>2</sub>B, Mn<sub>2</sub>B, Fe<sub>2</sub>B, Co<sub>2</sub>B, Ni<sub>2</sub>B, Mo<sub>2</sub>B and Cr<sub>2</sub>B borides as entropy, enthalpy, heat capacity and their temperature dependences.

This approach makes it possible to give more thorough from the thermodynamic point of view description of borides as well as monoborides formed from the liquid.

## Conclusions

In paper are considered structural and physical properties of the  $W_2B$ ,  $Mn_2B$ ,  $Fe_2B$ ,  $Co_2B$ ,  $Ni_2B$ ,  $Mo_2B$  and  $Cr_2B$  borides in binary alloys with boron weight content of 5.0-10.0 % (wt.), the rest is metal.

It should be noted that thermodynamic functions of the phases enable to predict physical and chemical properties of alloys under variable environmental conditions such as temperature, pressure and so on. The well-known computational techniques for thermodynamic functions of the phases can be applied only at equilibrium conditions and do not account for

fluctuation processes. Thereby, in the paper the temperature dependences of such thermodynamic functions as Gibbs energy, entropy, enthalpy and heat capacity  $C_p$  for the  $W_2B$ ,  $Mn_2B$ ,  $Fe_2B$ ,  $Co_2B$ ,  $Ni_2B$ ,  $Mo_2B$  and  $Cr_2B$  borides are obtained with accounting for the first degree approximation of high-temperature series of the free energy potential.

The obtained calculation results for thermodynamic functions of borides are in good agreement with experimental data.

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## Фізичні та термодинамічні властивості боридів

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У роботі досліджено фізичні властивості та термодинамічні функції боридів  $X_2B$  ( $X = W, Mo, Mn, Fe, Co, Ni$  та  $Cr$ ) з урахуванням флуктуаційних процесів. Для визначення фізичних властивостей сплавів використовували мікроструктурний, рентгеноструктурний та дюрOMETричний аналізи. В роботі було визначено фазовий склад сплавів та фізичні властивості боридів.

Вперше визначено термодинамічні функції боридів з використанням моделі Хіллера і Стеффансона та з урахуванням першого ступеня наближення високотемпературного розвинення термодинамічного потенціалу бінарних сплавів. Для боридів  $X_2B$  ( $X = W, Mo, Mn, Fe, Co, Ni$  та  $Cr$ ) отримано залежності від температури таких термодинамічних функцій, як енергія Гіббса, ентропія, ентальпія й теплоємність  $C_p$ , а також визначено їх значення при температурі утворення. Використаний у даній роботі підхід дає можливість надати найбільш повний з термодинамічної точки зору опис боридів, що утворюються з рідини.

Отримані результати розрахунків термодинамічних функцій боридів добре узгоджуються з експериментальними даними та даними інших авторів.

**Ключові слова:** бориди, енергія Гіббса, ентропія, ентальпія, теплоємність, флуктуаційний процес.